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Molecular-level insights on NIR-driven photocatalytic H₂ generation with ultrathin porous S-doped g-C₃N₄ nanosheets

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ABSTRACT

Unraveling how mid-gap state energy level of graphitic carbon nitride (g-C₃N₄) promote near-infrared (NIR) driven photochemical energy conversion at the molecular level remains a grand challenge. Here, we report a series of S double-site-doped ultrathin g-C₃N₄ nanosheets (SUCN) with adjustable intermediate band gap benefits from light response over NIR region. The SUCN produced after optimizing S double-site doping can effectively generate hydrogen (H₂) under NIR-light irradiation. The highest H₂ generation rate achieved was respectively 9.35 and 17.46 μ mol g⁻¹ h⁻¹ under λ = 765 and λ > 800 nm, which is firstly expended photocatalytic activity of S-doped g-C₃N₄ to NIR region beyond λ > 765 nm. We proposed a molecular-level method, i.e., the localized oxidation state of proton acceptor triethanolamine (TEOA) in the mid-gap state to ensure the NIR-driven H₂ generating behavior.

1. Introduction

Photocatalytic hydrogen evolution (PHE) using semiconductors and solar energy is a promising technology for the production of clean and renewable chemical fuels [1]. Of the various photocatalysts that have been reported for PHE, crystalline graphitic carbon nitride (g-C₃N₄) has garnered much attention as metal-free and high abundance [2]. Unfortunately, inadequate light utilization and severe photocarrier recombination are generally considered as typical constraints of g-C₃N₄ [3]. However, only a few portions of single-component g-C₃N₄ could meet expectations. [4]. Therefore, producing single-component g-C₃N₄ with high PHE performance under NIR light are capable as one of valuable challenges [5].

Recently, the covalent heteroatom-doping (N, C, B, S, etc.) photocatalysis technology coupled with g-C₃N₄ has attracted extensive research interest due to its expanding light response and increasing charge separation rate [6–11]. Covalent heteroatom-doping can be described as electron acceptors for photocarrier. Especially, the introduction of atom-sized S has been shown to not only maximize the activity of the C atom adjacent to the heteroatom, but also adjust the interlayer distances of g-C₃N₄ [12,13]. Moreover, the half-filled 3p orbital of S could induce an intermediate band gap in the intermediate band gap of g-C₃N₄ to accommodate more long-wavelength excited

charge carriers and promote the excitons separation [14,15]. However, although S-doping has made an important contribution to the local electric neutral state in the heptazine-conjugated skeleton, the mid-gap state energy level caused by the doping position was seldom investigated [16]. Controlling the S-doping sites in the heptazine-conjugated skeleton and elucidating the mechanism of mid-gap state energy level is considered to be a crucial challenge for circumventing NIR-energy conversion limitation [17,18].

Herein, we demonstrate the synthesis of S double-site-doped ultrathin g-C₃N₄ nanosheets (SUCN) via the in-situ synthesis strategy in the bottom-up acidification process. Compared to a trace amount of $\rm H_2$ evolution by bulk g-C₃N₄, the optimum $\rm H_2$ generation rate of SUCN reached 161.32 µmol h $^{-1}$ under the Vis light radiation. Unexpectedly, the optimum specimen also showed evolution rate of $\rm H_2$ (9.35 and 17.46 µmol g $^{-1}$ h $^{-1}$) under $\lambda=765$ and $\lambda>800$ nm, respectively. Urbach calculations confirmed that the SUCN were formed intermediate energy level positions between forbidden band. To get clear proofs of NIR-driven $\rm H_2$ evolution mechanism, the localized oxidation state of trie-thanolamine (TEOA) under the NIR irradiation was carried out. The results indicate that NIR-driven $\rm H_2$ generation is driven by excited electrons in the mid-gap state.

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2. Experimental section

2.1. Materials

Melamine (MA, \geq 99%), trithiocyanuric acid (TTCA, \geq 99%), acetonitrile (ACN, \geq 99%), H₂SO₄ (\geq 99%), ethyl alcohol (EtOH, \geq 99%), Na₂SO₄ (\geq 99%), triethanolamine (TEOA) and H₂PtCl₆ were purchased from Sinopharm Chemical Reagent Co., Ltd and all reagents employed in this study without further purification. Deionized water throughout the whole experimental process was purified by an ultrapure water system supplied by RSJ Co., Ltd.

2.2. Synthesis of photocatalysts

The SUCN was synthesized via a simple method with some optimizations. Briefly, 400 mg of MA, a certain amount of TTCA (90, 180, 270, 360, and 450 mg) and 3 mL $\rm H_2SO_4$ was added in 30 mL of ACN aqueous solution and stirred for 4 h. Then, transferred the suspension to the 50 mL Teflon-lined autoclave and heated at 100 °C for 24 h. The obtained precursor washed with excess EtOH and $\rm H_2O$ and dried under vacuum at 60 °C. After grounding into powder, SUCN was obtained by heated in Ar atmosphere at 550 °C for 4 h with a ramping rate of 7 °C min $^{-1}$. The SUCN was labeled x-SUCN (x = 0.09, 0.18, 0.27, 0.36, and 0.45 g), where x represented the quality of TTCA added into the precursor.

Meanwhile, a sample of $g\text{-}C_3N_4$ nanosheets was prepared similarly, except that 360 mg TTCA or 3 mL H_2SO_4 was not added [19,20]. The S doping $g\text{-}C_3N_4$ was labeled SCN and ultrathin $g\text{-}C_3N_4$ was labeled UCN [21].

2.3. Physicochemical characterization

X-ray diffraction (XRD) patterns were performed on a D/MAX-2500 diffractometer with Cu K α radiation ($\lambda = 1.54178$ Å) and radiation in the 2θ domain of 10.0-80° (7.0°/min). FT-IR was collected on the Nexus 470 (Thermo Electron Corporation). SEM was examined on an S-4800 field-emission scanning electron microscope (Hitachi, Japan). Transmission electron microscopy (TEM) was were obtained using the Tecnai G2 F30 S-TWIN. Elemental mapping pattern was recorded by an energydispersive/ X-ray spectrometer (EDS) attached to the TEM instrument. X-ray photoelectron spectroscopy (XPS) measurements were implemented by the Thermo Scientific ESCALAB 250Xi. UV-visible spectroscopy (UV-Vis) was characterized by a Shimadzu UV-2450 spectrometer. Photoluminescence (PL) spectra was operated on the Quanta MasterTM 40 fluorophotometer spectrometer. The solid-state ¹³C nuclear magnetic resonance (13C NMR) spectra was investigated on a Switzerland Bruker model 600 M spectrometer by Shiyanjia Lab (www. shiyanjia.com). Time-resolved photoluminescence spectra was recorded on a Quanta Master & Time Master Spectro-fluorometer. All photoelectrochemical measurements were measured by using an electrochemical workstation (CHI-852 C, Shanghai Chenhua Limited, China) with a conventional standard three-electrode cell. All photoelectrochemical measurements were carried out in a CHI-852 C electrochemical workstation.

2.4. Photocatalytic reaction

Photocatalytic activities were evaluated in quartz reactor by dispersing the catalysts (20 mg) in a 50 mL aqueous TEOA solutions (10 vol%). The reaction mixture was irradiated by a 300 W xenon lamp (100 mW cm $^{-2}$) after loading on 2 wt% Pt with $\rm H_2PtCl_6\cdot 6~H_2O$ aqueous solution. The temperature was maintained at 5 °C by a flow of cooling water throughout the photocatalytic reaction process. The amount of $\rm H_2$ generation rate was evaluated on a GC-7920 gas chromatograph every hour. Besides, the cycle test and the apparent quantum efficiency (AQE) of photocatalytic $\rm H_2$ production rate was carried out under the same conditions.

2.5. Computational details

All the spin-polarized density functional theory (DFT) were consummated using the Vienna ab initio simulation package (VASP). In order to get accurate results, the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional within a generalized gradient approximation and the projector augmented-wave potential were calculated. The heteroatoms of S were embedded in a $2\times2\times1$ octahedron the bulk g-C_3N_4 with a vacuum of 20 Å. The K point division $1\times1\times1$ was sampled in a $3\times3\times1$ Monkhorst-Pack grid to structural optimization and self-consistent calculation. The energy cutoff was set to 550 eV for the plane-wave expansion.

3. Results and discussion

3.1. Morphological characterization

The possible polymerization process for incorporating off S building blocks from TTCA into the heptazine-conjugated skeleton of g-C₃N₄ structure was fabricated (Fig. 1. a). The thiol group of thiocyanuric acid reacts with the hydroxyl group in water to form a thioether bond. Concentrated sulfuric acid dissolved in acetonitrile oxidizes the thioether bond to form matte salt prepolymer. At the same time, as Brønsted acid, concentrated sulfuric acid breaks the hydrogen bond between melamine plates. The XRD patterns confirmed the similar feature of SUCN. Moreover, the (002) peak in Fig. S1. (a) provided a significant reduction in the interlayer distance of other contrast samples [22]. The XRD patterns of bulk g-C₃N₄ in Fig. S1. (a) showed two peaks at 13.2° and 27.1° that was attributed to crystal face characteristics of interlayer repeating structural and the in-plane stacking of aromatic segments, respectively. Although the x-SUCN showed similar crystal structures to g-C₃N₄, the. width of (002) peak shape became broadens and up-shifted. This result indicated that a synergy effect of S double-site-doped and acid etching reduced the crystallinity and destroyed the orderly arrangement of g-C₃N₄ [16].

The crystal phase and microstructure of g-C₃N₄ and S-doping g-C₃N₄, were studied by SEM and TEM. As displayed in Fig. 2. (a-b), the g-C₃N₄ showed a representative stacked structure, while the 0.36-SUCN displayed porous multilayer two-dimension nanosheets with irregular nanopores (5-25 nm), which were consistent with Fig. S1. (b-d) in the Supporting Information. All the elements (i.e., N, C, and S) were homogeneous distributed on the 0.36-SUCN in Fig. 2. (c-f), suggesting that S has been incorporated into the heptazine-conjugated skeleton [23]. The AFM in Fig. 2. (g) demonstrated that 0.36-SUCN was the laminar structure with ultrathin thickness. The stochastically measured thickness of the nanosheets was 1.1 nm in Fig. 2. (h). The N2 sorption isotherms of 0.36-SUCN and g-C₃N₄ exhibited a typical type-IV curve with a representative H₃-type hysteresis loop in Fig. 2. (i) [24]. The specific surface area (SSA) of 0.36-SUCN based on BET was calculated to 38.21 m² g⁻¹ with a nearly 3-fold increasement than that g-C₃N₄ $(12.96 \text{ m}^2 \text{ g}^{-1})$. As exhibited in Fig. S1. (f), the pore volume obtained from the pore size distribution curve of 0.36-SUCN and bulk g-C₃N₄ were $0.26~\mathrm{cm^2~g^{-1}}$ and $0.14~\mathrm{cm^2~g^{-1}}$, respectively. Thus, the highly micro mesopore structure of 0.36-SUCN not only shorten the diffusion path of photoexcited carriers, but also supplied a larger surface area for photocatalytic active sites [21,25].

3.2. Structure characterization and optical/electronic properties

Molecular functional groups of a series of photocatalysts were collected by FT-IR [26]. The broad absorption characteristic peaks located in 1100–1650 cm $^{-1}$ were assigned to the C-N skeletal vibration in Fig. 3. (a) & S1. (b) [3]. The FTIR spectra of the S-doped g-C₃N₄ indicated only the vibration peak relatively changes at 808 cm $^{-1}$ owing to the low S content [27]. The XPS S 2p spectra (Fig. 3. (b)) to further confirm the presence of sulfur in 0.36-SUCN material. The signal

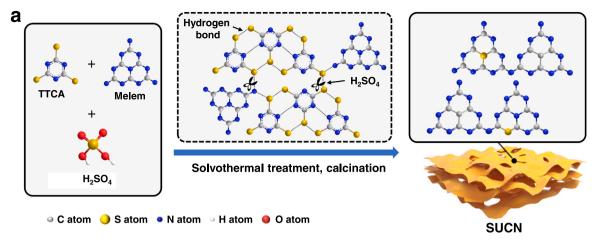


Fig. 1. (a) Schematic representation for the SUCN synthesis process.

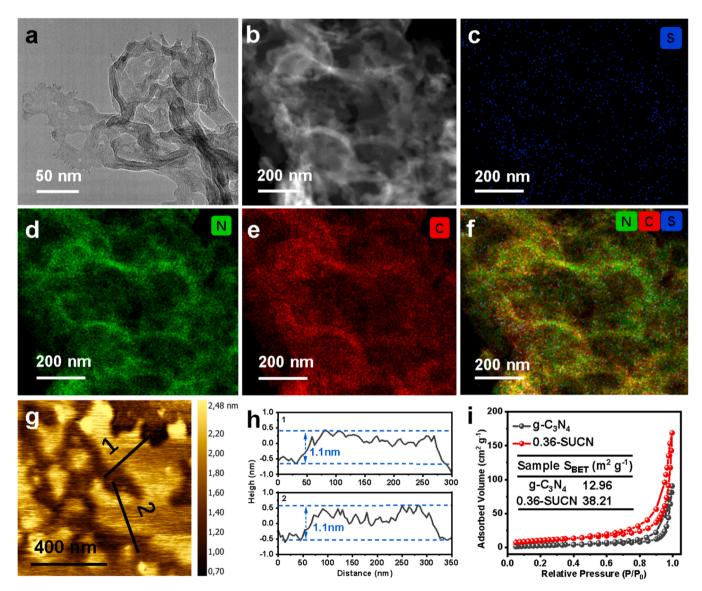


Fig. 2. (a) TEM image of 0.36-SUCN (b) the inset of TEM image showed the corresponding region for mapping; (c-f) EDS mapping of 0.36-SUCN. (g) AFM image and (h) corresponding height profiles along the white line in d of 0.36-SUCN. (i) N_2 sorption isotherms of g-C₃N₄ and 0.36-SUCN.

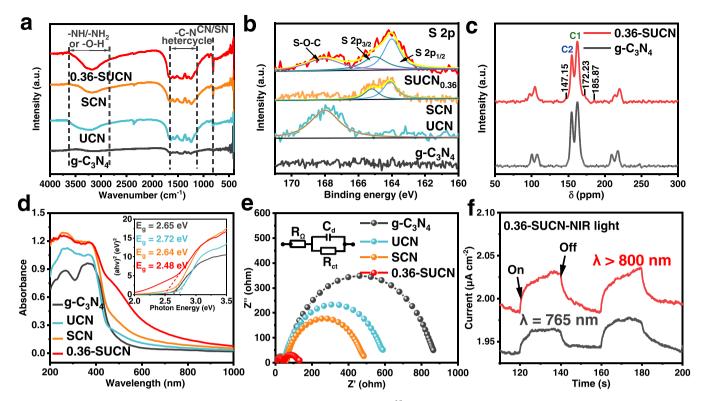


Fig. 3. (a) The FTIR spectra. (b) High-resolution S 2p XPS spectrum. (c) The solid-state ¹³C NMR spectra. (d) UV-visible plots. Inset: the relationship between Kubelka-Munk function and photon energy. (e) Impedance spectra. Inset: equivalent circuit impedance model. (f) Transient photocurrent curves under NIR light.

intensity of S 2p was not strong (Fig. S3 (a)) due to the low doping content. The binding energy of S 2p was deconvoluted into four peaks, which were located at 168.1, 165.0, and 164.0 eV. The peaks at 164.0 and 165.0 eV were assigned to the S 2 $P_{1/2}$ and S 2 $P_{3/2}$ bonds, while another weak peak at 168.1 eV were attributed to the high-valent S species formed during calcination [28]. Moreover, the variations of atomic ratios between g-C₃N₄ and 0.36-SUCN (Table S1) were also demonstrated by XPS. The N/C atomic ratio of 0.36-SUCN decreased from 1.293 to 1.229. Additionally, the N/C and S/C ratios of 0.36-SUCN is much lower, indicating the existence of nitrogen defects [29]. Simultaneously, the S element can act as a photoelectron capturing site, which greatly reduces the backward charge recombination, and ultimately benefits the photo-redox process [30].

Moreover, the chemical structure of 0.36-SUCN was further investigated via the solid-state ¹³C NMR in Fig. 3. (c) & S3. (d). Typical, ¹³C NMR peaks of the bulk g-C₃N₄ (151 MHz, none) were 217.68, 209.48, 162.61, 154.60, 107.65 and 100.28. In contrast, 0.36-SUCN exhibited three distinct peaks [13 C NMR (151 MHz, none) δ 220.03, 213.83, 185.87, 172.23, 162.58, 154.87, 147.15, 104.47, and 97.34.]. The two weak peaks located at 185.87 and 147.15 ppm were probably attributed to the chemical shifts of the C3 atom ($CN_2 = S$) and C4 atom ($CN_2 = N$), respectively in Fig. S4. Another weak peak at 172.23 ppm was probably related to C5 atoms of the S-CN2 group. The UV-Vis DRS was implemented the optical characteristics of g-C₃N₄ and other contrast samples in Fig. 3. (d) & S3. (e). The optical absorption of 0.36-SUCN was greatly enhanced in Fig. S3 (e), indicating the reduced band gap [31]. As expected, 0.36-SUCN showed an obvious absorption tail over 400-700 nm and extended to the longer wavelength (near 1000 nm). The energy gap (Eg) of bulk C3N4 acquired by the Kubelka-Munk-plot transform function was 2.65 eV [32]. In comparison, the obtained decreasing trend of E_g valued from 2.65 to 2.48 eV for x-SUCN.

Apart from the suitable E_g , the relevant energy level matching the redox potential reaction was also a major parameter evaluate the performance of photocatalysts [33]. On the basis of the Mott-Schottky plots in Fig. S5. (a-h), the samples displayed positive slops, demonstrating the

n-type character of g-C₃N₄ [34]. The flat band positions (E_{FB}) of the g-C₃N₄ was -1.087 V (vs Ag/ AgCl), while 0.36-SUCN was -1.167 V. Generally, the CB potential of n-type semiconductors was approximately 0.1 V higher than E_{FB} , thus the CB of g-C₃N₄ and 0.36-SUCN were -0.987 eV and -1.067 eV, respectively [35]. The 0.36-SUCN showed a fast and forceful photocurrent response in Fig. S3 (f), and such photocurrent response phenomenon has been acceptive repeated in second on-off cycles. Obviously, the photocurrent response curves of 0.36-SUCN were still clearly observable under NIR illumination in Fig. 3. (f).

The photoinduced excitation splitting properties of the photocatalyst was evaluated by PL emission spectroscopy. As exhibited in Fig. S3. (g), the PL intensity of 0.36-SUCN was lowest [17,18]. In Fig. 3. (e) & S3. (h), the arc radius of the semicircular Nyquist curve from electrochemical impedance spectroscopy (EIS) decreased gradually. This was attributed to the effective shuttle from the electrode to electrolyte and the rapid transfer of interfacial charge on a hydrophilic surface. The lowest EIS of 0.36-SUCN indicated that the ultrathin layer morphology and S-doping in the carbon nitride framework can greatly enhance charge transport (Fig. S3. (h)). Besides, the average fluorescence lifetimes are 3.834 and 2.223 ns for bulk g-C₃N₄ and 0.36-SUCN (the detailed information is listed in Fig. S3. (i) & Table S2, which was consistent with our assumption. The rapid and uniform photocurrent response curves (Fig. S5. (i)) demonstrated that the x-SUCN possessed a significant higher photocurrent density, confirming the increase of photogenerated charge. Especially, the highest charge separation efficiency of 0.36-SUCN among the x-SUCN was attributed to the ultrathin characteristics and S-doping, which obtains similar trends in EIS and PL measurements.

3.3. Photocatalytic performance

The PHE activity of x-SUCN were measured under visible light. As shown in Fig. 4. (a) and S6. (a-b), g-C₃N₄ showed poor H₂ generation activity under irradiation of $\lambda > 420$ nm, which could attribute to the ultra-high electron-hole recombination rate. All-modified

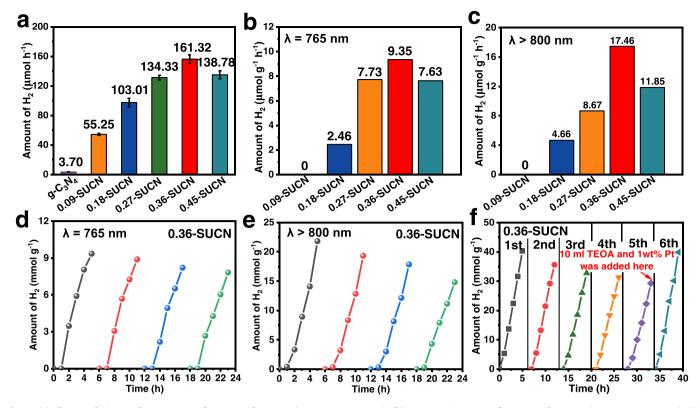


Fig. 4. (a) Photocatalytic H_2 release amount of g-C₃N₄ and x-SUCN (λ > 420 nm, pH = 7). (b) H_2 generation rates of g-C₃N₄ and x-SUCN (λ = 765 nm, pH = 7). (c) H_2 generation rates of g-C₃N₄ and x-SUCN (λ > 800 nm, pH = 7). (d-f) H_2 generation stability estimation of 0.36-SUCN.

photocatalysts exhibited noticeably enhanced activity relative. Surprisingly, 0.36-SUCN showed the highest $\rm H_2$ generation rate (161.32 mmol $\rm h^{-1}$), which is about 43 times higher than that of g-C₃N₄ (3.701 mmol $\rm h^{-1}$). The PHE of 0.36-SUCN is significant, and is significantly better than most reported single-element doped and functionalized g-C₃N₄ in Table S3.

The results of PHE under NIR light irradiation were measured in Fig. 4. (b-c). Obviously, the x-SUCN has enhanced PHE activity under NIR light. The above data demonstrated that S-doping g-C₃N₄ porous multilayer nanosheet could broaden the light to NIR for the first time. In addition, the stability performance of 0.36-SUCN was collected under the same conditions. Moreover, the PHE of 0.36-SUCN was stable under $\lambda = 765$ nm and $\lambda > 800$ nm (Fig. 4. (d & e)). As presented in Fig. 4. (f), H₂ evolution reaction of 36-SUCN slightly decreased during the first five cycles, which might be attributed to the consumption of sacrificial agent and cocatalyst. To prove this regard, we added 10 mL TEOA and 1 wt% H₂PtCl₆·6 H₂O before the start of the sixth hydrogen production cycle. There was no significant discrimination between the sixth H₂ production cycle and the first H₂ production cycle, suggesting the outstanding stability of 0.36-SUCN under the applied reaction conditions. Moreover, the XRD pattern of the 0.36-SUCN has unchanged after six photocatalytic reactions in Fig. S6. (d). The AQE performances of 0.36-SUCN were measured under a series of wavelength irradiation for exploring wavelength-dependent H2 evolution. It can be noticed that the AQEs of 0.36-SUCN at 400 and 420 nm were estimated to be about 22.91% and 17.75%, respectively in Fig. S6. (c). Since the band tail absorption was not completely uniform, AQE cannot show a theoretical linear correlation [26].

3.4. Theoretical calculations

The PHE of SUCN was closely interrelated to the electronic structure of S in SUCN sites. The values of work function of SUCN and g- C_3N_4 was calculated by DFT in Fig. 5. (a-b). The results showed that the energy

required for electrons to escape from Fermi level of the SUCN to the vacuum level is much smaller than that of the g-C₃N₄. Combined with Fig. 5. (c) & (d), the S 3p hybridizes adjacent atoms presented an electron-rich environment, which was consistent with the electron donor [36]. Densities of states (DOS) was also studied (Fig. S7), considering that the CB and VB of g-C₃N₄ were both composed of C 2p and N 2p. Thus, the total DOS showed that the band gaps of g-C₃N₄ was ca. 1.02 eV. In addition, there was no impurity peak between the band gaps. Since the calculation of the exchange correlation function underestimates the energy gap for its own limitations. For SUCN, S 3p participated in the edge hybridization of the valence band, so the CB and VB were composed of C 2p, N 2p and S 3p. As a result, S doping narrowed the band gap and induced a red-shift on the absorption edge. A new impurity peak appeared between the band gaps due to S 3p via the analysis wave state. The appearance of impurity peaks indicated S-doping forms a new energy level between forbidden band and induces $n\rightarrow\pi$ * electronic transition, widened the light absorption range and further enhanced the PHE performance [37].

3.5. The mechanism of the photocatalytic reaction

According to Urbach formula, the band edge of the intermediate gap state could be roughly displayed in combination with the constant VB position in Fig. S8. [38,39]. The intermediate state of 0.36-SUCN was close to the CB edge, which shortened the transition distance from the intermediate state to CB edge. Carriers were more easily excited intermediate gap states, and the gap can accommodate electrons excited by longer wavelength photons [40]. Therefore, the CB and VB energy levels could be defined by combination of the CB edges with the Eg values. The band structures could be inferred as Fig. 6. (a) and S8. (e).

A schematic diagram of the energy band and intermediate gap structure is illustrated in Fig. 6. (b), and the possible process of H_2 release under Vis and NIR light is proposed:

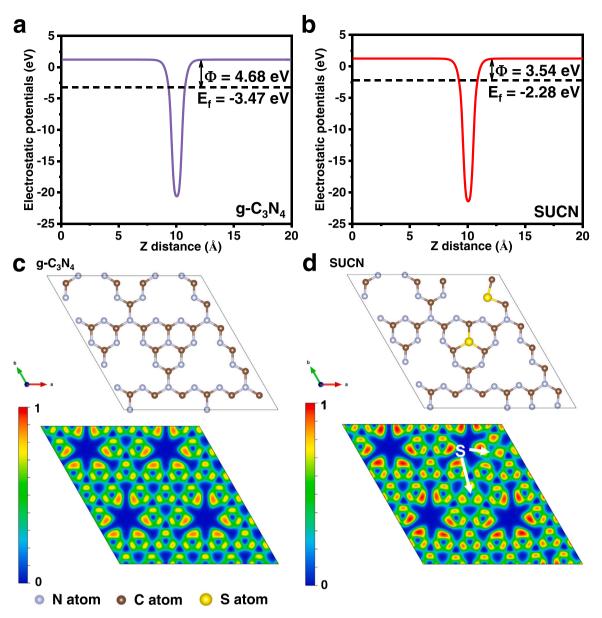


Fig. 5. (a-b) The calculated work functions of g-C₃N₄ and SUCN. The DFT calculations of the electronic structure of (c) g-C₃N₄ and (d) SUCN.

$$0.36 - SUCN + h\nu \rightarrow e_{0.36-SUCN}^{-} + h_{0.36-SUCN}^{+}$$
 (1)

$$TEOA + h_{0.36-SUCN}^{+} \rightarrow TEOA^{+}$$
 (2)

$$2H^{+} + 2e_{0.36-SUCN}^{-} \rightarrow H_{2}$$
 (3)

The local photooxidation of TEOA during photocatalytic simulation reveals the behavior of $\rm H_2$ generation driven by the intermediate band gap under the NIR light. The valence band of 0.36-SUCN showed good oxidation behavior, thus an ethyl oxide peak appears under Vis light in Fig. 6. (c). There was almost no difference with TEOA raw material under NIR radiation, because the intermediate band gap is more negative than the valence band of 0.36-SUCN. In view of above experimental data and DFT results, the enhanced PHE activity of SUCN could be ascribed to three points: Firstly, S-doping formed a suitable intermediate band gap between CB and VB to capture long-wavelength excited e. Second, S-doped in the triazine skeleton acted as an electron acceptor could reduce the recombination rate of e h⁺ pairs. Finally, the porous nanosheet structure provided a large specific surface area, which could capture numerous photons and shorten the distance of e transferring from the inside to the surface of the photocatalyst. In addition, porous

structures could provide marginal active sites. Therefore, 0.36-SUCN showed broadband gap absorption, ultra-high visible PHE performance and near-infrared PHE.

4. Conclusions

In conclusion, the S-doped g-C₃N₄ porous ultrathin material was prepared by a simple heat treatment method. As a result, by doping the triazine skeleton with S element, SUCN could accelerate the separation and transfer of charges, showing excellent H₂ generation activity and high solar energy availability. The PHE of the best sample was 43 times higher than that of the bulk g-C₃N₄ under $\lambda >$ 420 nm radiation. The S-doped g-C₃N₄ firstly produced H₂ under NIR light, and its PHE under $\lambda =$ 765 and $\lambda >$ 800 nm radiation was 9.35 and 17.46 μ mol g $^{-1}$ h $^{-1}$, respectively. The significant relationship between molecular configuration and charge dynamics was demonstrated by DFT calculations. In particular, the PHE performance under NIR radiation was proved by a molecular-level method to be driven by the appropriate intermediate band gap provided by the S-electron acceptor. The precise regulation of element doped g-C₃N₄ affords a new strategy for greatly improving the

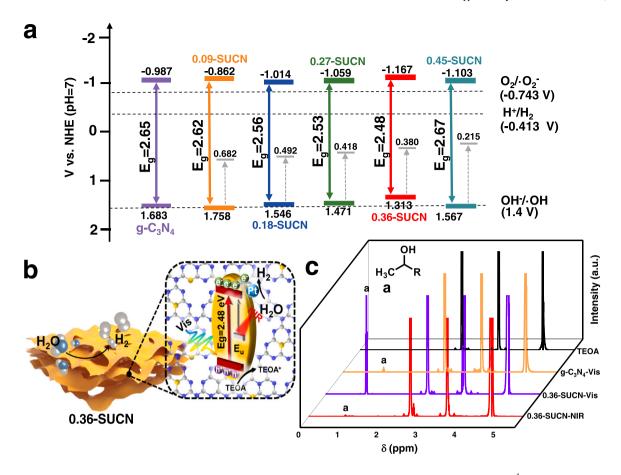


Fig. 6. (a) The schematic diagram of the energy band structure. (b) Schematic diagram of H_2 generation through 0.36-SUCN. (c) ¹H NMR of spectroscopy of TEOA after 5 h irradiation in D_2O .

broad-spectrum solar-energy utilization efficiency.

CRediT authorship contribution statement

Xiaojie Wu: Conceptualization, Data curation, Formal analysis, Supervision, Validation, Visualization, Writing – review & editing. Di Li: Review & editing, Data curation, Supervision. Bifu Luo: Data curation, Supervision. Biyi Chen: Visualization, Investigation. Yuanyong Huang: Validation, Visualization. Nanjun Shen: Investigation. Tingting Yu: Investigation. Longhua Li: Investigation, Supervision. Weidong Shi: Funding acquisition, Investigation, Methodology, Project administration.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2022.122292.

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